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# Free carrier generation and recombination in PbS quantum dot solar cells

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Time Delayed Collection Field and Bias Assisted Charge Extraction (BACE) experiments are used to investigate the charge carrier dynamics in PbS colloidal quantum dot solar cells. We find that the free charge carrier creation is slightly field dependent, thus providing an upper limit to the fill factor. The BACE measurements reveal a rather high effective mobility of  $2 \times 10^{-3} \text{ cm}^2/\text{Vs}$ , meaning that charge extraction is efficient. On the other hand, a rather high steady state non-geminate recombination coefficient of  $3 \times 10^{-10} \text{ cm}^3/\text{s}$  is measured. We, therefore, propose a rapid free charge recombination to constitute the main origin for the limited efficiency of the PbS colloidal quantum dots cells. © 2016 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4943379>]

Colloidal quantum dots (CQDs) are a promising class of materials for optoelectronic applications due to their well-tunable optical and electronic properties. In the recent years, improvement in the quality of the CQDs, as a consequence of better and cleaner synthetic processes, has stimulated their use in a variety of applications from light emitting diodes<sup>1</sup> to solar cells,<sup>2,3</sup> field effect transistors,<sup>4-6</sup> and photocatalytic devices.<sup>7,8</sup> The size-induced quantum confinement in the quantum dots can be tuned via precise tuning of the QDs dimensions. This quantum confinement can be partially maintained even in strongly coupled CQD solids.<sup>9</sup> Strong coupling is achieved through the ligand exchange process, by which the original, bulky organic ligands are replaced with shorter ones, enhancing the electronic coupling. Lead chalcogenide quantum dot solids exhibit a very broad absorption range, extending into the near infrared region.<sup>10</sup> This makes them particularly attractive for photovoltaic devices with high photocurrents.<sup>11</sup> From the early Schottky devices,<sup>9,11</sup> which show limited open circuit voltage ( $V_{oc}$ ) and fill factors ( $FF$ ), great improvement in the device structure and in the choice of ligands led to a record PbS CQD solar cell with efficiencies up to 10%.<sup>12</sup> Still, the device performance seems to be limited by the  $V_{oc}$  and the  $FF$ , purportedly due to recombination assisted by a large number of traps<sup>13</sup> or geminate recombination.<sup>14</sup> The nature of traps and the recombination mechanism in the CQD solar cells is not yet fully understood.

To date, most works have focused on the transport in CQD solids<sup>15-17</sup> or the photophysics,<sup>18</sup> and only a few have investigated the working mechanisms of CQD solar cells

in detail.<sup>14,19,20</sup> In this letter, we use the time-delayed collection field (TDCF),<sup>21</sup> bias-assisted charge extraction (BACE),<sup>22</sup> and steady state measurements to investigate free charge generation and recombination in PbS CQDs Schottky structure solar cells. We find the charge generation to be slightly field-dependent, the charge extraction to be reasonably efficient, and the free charge loss to be dominated by bimolecular recombination.

The Schottky-junction solar cells based on 3.8 nm PbS CQDs were fabricated following the procedure described in the Experimental section. The  $JV$ -curve and parameters of the device investigated here are shown in supplementary Figure S1.<sup>23</sup> BACE was utilized to measure the carrier density in the devices at  $V_{oc}$  under different illumination conditions. In BACE, the sample is illuminated with a constant light intensity for ca. 3 ms to reach the steady state. Then, the illumination is switched off, and a reverse bias is applied to extract all mobile carriers from the active layer. A blue 445 nm cw-laser diode was used as a light source. To avoid carrier losses due to the free carrier recombination, the illumination was switched-off within 10 ns while the external bias was simultaneously switched from  $V_{oc}$  to the collection bias  $V_{coll}$ . Figure 1(b) shows the resulting carrier density as a function of the steady state light intensity.

At  $V_{oc}$ , the net current in the device is zero, meaning that all the photogenerated carriers must recombine. If the generation rate  $G$  is proportional to light intensity and independent of field, the extracted charge density  $n$  as a function of illumination intensity at the respective  $V_{oc}$  yields the order of recombination via

$$n = \left( \frac{G}{k_\alpha} \right)^{\frac{1}{\alpha}}. \quad (1)$$

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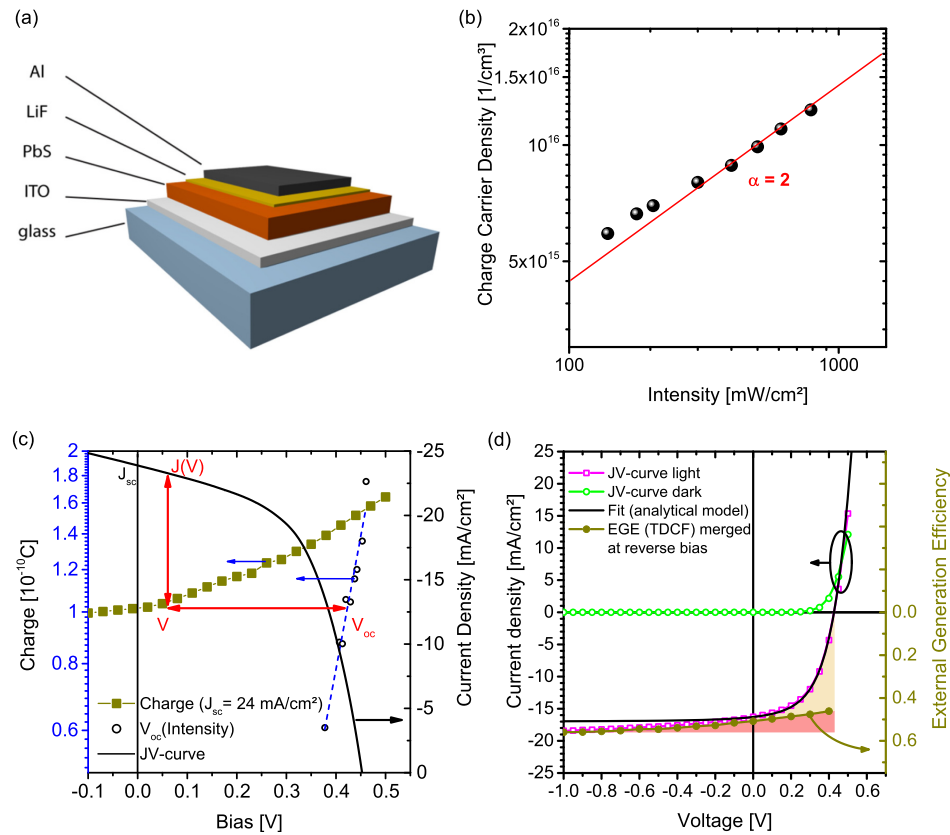


FIG. 1. (a) Schematic device structure, the PbS QDs are cross-linked with BDT. (b) Charge carrier density measured with BACE at different illumination intensities at a bias which equals  $V_{oc}$  (black circles), with the red line showing the slope for strict bimolecular recombination, with a recombination coefficient  $k_2 \sim 3 \times 10^{-10} \text{ cm}^3/\text{s}$ . (c) The BACE measurements of the steady state carrier density at different illumination conditions in the operating voltage range. Shown are measurements with increasing intensity at  $V_{oc}$  (black circles) and at a fixed intensity (yielding a short circuit current of  $24 \text{ mA}/\text{cm}^2$ , dark yellow squares). The corresponding  $JV$ -curve is shown by a black solid line.  $V$  is the bias where the charge carrier density  $n$  corresponding to the current  $J(V)$  was obtained. (d) Bias dependent current in the dark (open green circles) and at AM1.5G (pink open squares), compared to the field dependence of free charge generation measured with TDCF (wavelength 532 nm, delay time 4.5 ns, and dark yellow circles). The area above the measured EGE shows non-geminate and the area below geminate losses. The solid line is the prediction from an analytical model based on experimentally determined parameters.

The data in Figure 1(b), plotted in a log-log-fashion, approximate a slope of  $1/2$  for higher illumination intensities, in accordance with a strict bimolecular recombination process (recombination order  $\alpha=2$ ) with a density independent free carrier recombination coefficient  $k_2$ . At the lower illumination intensities, the extracted charge carrier density is expected to deviate from the true average carrier density in the active layer, due to an inhomogeneous charge distribution.<sup>24</sup> Therefore, the results for illumination intensities below ca. 1 sun ( $100 \text{ mW}/\text{cm}^2$ ) have been omitted. To determine the absolute value of  $k_2$ , the generation rate was estimated from the saturated reverse photocurrent  $J_{sat}$ . With  $J_{sat} = 20 \text{ mA}/\text{cm}^2$  at an illumination intensity of AM1.5G at  $100 \text{ mW}/\text{cm}^2$ , a reverse bias of  $-1.5 \text{ V}$ , and an active layer thickness  $d$  of  $140 \text{ nm}$ ,  $k_2 = 3 \times 10^{-10} \text{ cm}^3/\text{s}$ . This value is significantly larger than for most organic donor-acceptor blends, for which  $k_2$  is typically between  $10^{-12}$  and  $10^{-11} \text{ cm}^3/\text{s}$ .<sup>21,25–27</sup> As a consequence of rapid free carrier recombination, the charge density in the device extrapolated to 1 sun illumination intensity is fairly small, around  $n = 5 \times 10^{15} \text{ cm}^{-3}$ , while the polymer-based bulk heterojunction cells typically exhibit  $n$  in excess of  $10^{16} \text{ cm}^{-3}$ .<sup>22,28,29</sup>

As stated, at  $V_{oc}$ , all photogenerated carriers must undergo recombination. Thus, the losses in voltage from the bandgap to  $V_{oc}$  are due to recombination of some form or another. We propose that this rapid free carrier recombination

and the resulting low steady state carrier density at an illumination intensity of 1 sun is a major reason for the low  $V_{oc}$  and underperformance of these cells.

Despite the rather high coefficient for the free carrier recombination, our devices exhibit a reasonably high FF, meaning that the photogenerated charges are rapidly extracted from the active layer, even at low internal fields. This is perhaps surprising as the recombination and charge extraction are in competition with each other within the operating regime of a solar cell. Thus, the high FFs imply efficient charge extraction despite strong recombination, which suggests high charge carrier mobilities. Balazs *et al.*<sup>30</sup> reported impressively high field effect transistor (FET) electron mobilities  $\sim 10^{-2} \text{ cm}^2/\text{Vs}$  in colloidal quantum dot solids, but the hole mobilities were rather low,  $\sim 10^{-4} \text{ cm}^2/\text{Vs}$ . The mobilities for CQDs with BDT ligand, measured in field effect transistors by Bisri *et al.*, were significantly lower,<sup>31</sup> but these FET mobilities were shown to be strongly affected by traps at the interface between the CQD layer and the gate dielectric<sup>31</sup> and could be improved by 5 orders of magnitude using an ion gel gating.<sup>31</sup> Thus, the mobilities in the FET architecture are not necessarily applicable for understanding the solar cell operation.

We have recently proposed an approach to determine the effective charge carrier mobility  $\mu_{eff} = \frac{2\mu_e\mu_h}{\mu_e+\mu_h}$  within a solar cell at working conditions, which relies on the measurement of the  $V_{oc}$  and  $JV$  characteristics at varying charge



carrier densities. To be more specific, a certain carrier density  $n$  is established in two ways: either by illumination at intensity  $I_1$  and  $V_{OC}$  conditions, where the net-current density is zero or by keeping the device at a lower bias  $V < V_{OC}$ , but at higher illumination intensity  $I_2$ , with a measurable net-current  $J(V)$ . With these data, the effective mobility is calculated via<sup>29</sup>

$$\mu_{eff}(n, V) = \frac{J(V)d}{2en(V)(V_{oc} - V)}. \quad (2)$$

Figure 1(c) summarizes the values for the steady state carrier densities at the open circuit,  $n(V_{oc})$ , measured with increasing illumination intensity, together with  $n(V)$  for a fixed illumination intensity corresponding to a short circuit current of 24 mA/cm<sup>2</sup>.  $n(V_{oc})$  increases exponentially with  $V_{oc}$  at increased intensities, as expected and shown before.<sup>22,29</sup> In contrast, at a given intensity,  $n(V)$  varies more smoothly with bias. By definition,  $n(V)$  intersects with  $n(V_{oc})$  at the particular  $V_{oc}$ , but it gradually decreases when the bias is below  $V_{oc}$  due to a more rapid extraction of the carriers. In the working range of solar cells,  $\mu_{eff}$ , as derived from these data with Eq. (2), varies between  $1.5$  and  $3 \times 10^{-3}$  cm<sup>2</sup>/Vs, slightly depending on the bias and illumination intensity. The mobility, influenced by both the holes and electrons, compares well to that of some high FF organic solar cells.<sup>22,29</sup> For comparison, we have estimated the mobility of the faster carrier type from the initial decay of the photocurrent transients (see Figure S2), which yielded a value of  $\mu_{TDCF} = 8 \times 10^{-3}$  cm<sup>2</sup>/Vs. In view of the results by Bisri *et al.* discussed above, we assign this to the motion of electrons. As  $\mu_{eff}$  is markedly lower than  $\mu_{TDCF}$ , we conclude that the extraction of photogenerated charges out of the active layer is limited by the slow motion of holes.

In addition to issues with charge extraction, the FF can be limited by the field dependence of charge generation. To check for such dependence, the TDCF experiments were conducted as a function of excitation wavelength and bias. The experiments were performed with the excitation wavelengths of 410 and 532 nm, using a delay of only 4.5 ns between the pulsed illumination and initiation of charge extraction with an applied  $V_{coll}$  of  $-1.5$  V. This delay was chosen to ensure that the free carrier recombination had not yet commenced. For these conditions, we found the extracted charge to scale strictly linearly over a wide range of fluences (between  $0.005 \mu\text{J}/\text{cm}^2$  and  $0.7 \mu\text{J}/\text{cm}^2$ ), ruling out any significant losses by higher order processes (Figure S3). In the following, the efficiency of generation (EGE), expressed by the total extractable charge  $Q_{tot}$  in relation to the number of incident photons, is discussed at a rather low fluence (of only  $0.15 \mu\text{J}/\text{cm}^2$ ), as this yielded a charge carrier density comparable to the steady state conditions under 1 sun. Charge generation with the excitation wavelength of 532 nm was found to have a slight field dependence, which becomes somewhat stronger if the sample is instead excited at 410 nm (Figure S3). An EGE of 0.5 at the short circuit conditions indicates that 50% of the incident photons yield collected charges, and that at reverse bias, this improves to almost 55% as the stronger electric field suppresses geminate recombination. The field dependent external generation data are overlaid on the steady state  $JV$  characteristics in Figure 1(d). It becomes clear

that when driving the device at reverse bias, the increasing photocurrent can be directly attributed to the improved generation efficiency at these stronger fields, and reduced free carrier recombination. However, as the device approaches  $V_{oc}$ , the current becomes more strongly affected by the internal electrical field, which cannot be ascribed entirely to generation, but rather is attributed to the free carrier recombination (the light area in Figure 1(d)).

To check for consistency between the data gained by the BACE and steady state  $JV$  measurements, in Figure 1(d), the experimental  $JV$ -curve is compared to an analytical model based on the second order recombination, as published recently by Würfel and Neher.<sup>32</sup> Here, the current density  $J$  and the external bias  $V_{ext}$  are written as functions of the internal voltage  $V_{int}$ , where  $V_{int}$  is a direct measure of the quasi-Fermi level splitting in the bulk of the photoactive material

$$J(V_{int}) = J_0 \left\{ \exp \left[ \frac{eV_{int}}{k_B T} \right] - 1 \right\} - J_G, \quad (3a)$$

$$V_{ext}(V_{int}) = V_{int} + \frac{d}{2e\mu_{eff} \exp \left[ \frac{eV_{int}}{2k_B T} \right]} J(V_{int}). \quad (3b)$$

Here,  $J_G$  is the generation current,  $k_B$  is the Boltzmann's constant,  $T$  is the temperature, and  $J_0 = edk_2 n_i^2$  is the saturation current density, which depends on  $e$  is the elementary charge,  $k_2$  is the recombination rate, and  $n_i$  is the intrinsic charge carrier density. To calculate  $J(V)$ ,  $k_2$  and  $\mu_{eff}$  were taken directly from the BACE measurements,  $n_i$  was taken from extrapolating the high charge carrier density range measured with BACE to  $V_{oc} = 0$  V (see Figure S4 and Table S1), and  $J_G$  was set equal to the short circuit current (values are given in Table S1). Note that this approach does not implement the field dependent charge generation. Despite the simplicity of the approach, a very good agreement between the measured  $JV$ -curve and the prediction by the analytical model is seen (solid black line in Figure 1(d)), using the measured parameters (see Table S1). The model works particularly well at the forward bias, where the  $JV$ -curve is mainly governed by the non-geminate recombination.

TDCF experiments with longer delays afford a detailed look at the free charge dynamics following a pulsed excitation. Figure 2(a) displays the three relevant quantities, the charge flowing out of the device during pre-bias,  $Q_{pre}$ , the charge collected upon application of the reverse collection voltage,  $Q_{coll}$ , and the sum,  $Q_{tot}$ , plotted as function of delay time. Data were assembled for three different fluences, and all quantities were normalized to the total collected charge at the shortest delay time. Also, before each series of measurements, the charges present in the dark at the given pre-bias  $Q_{dark}$  were measured. A more detailed description of this technique and analysis can be found elsewhere.<sup>21</sup> With increasing delay time,  $Q_{pre}$  increases continuously, as more and more charge carriers leave the device before the collection bias is applied. On the other hand,  $Q_{coll}$  exhibits a continuous decrease, as the charge left in the device after  $t_{del}$  becomes diminished by both the extraction and non-geminate recombination. Thus, the drop of  $Q_{tot}$  with the

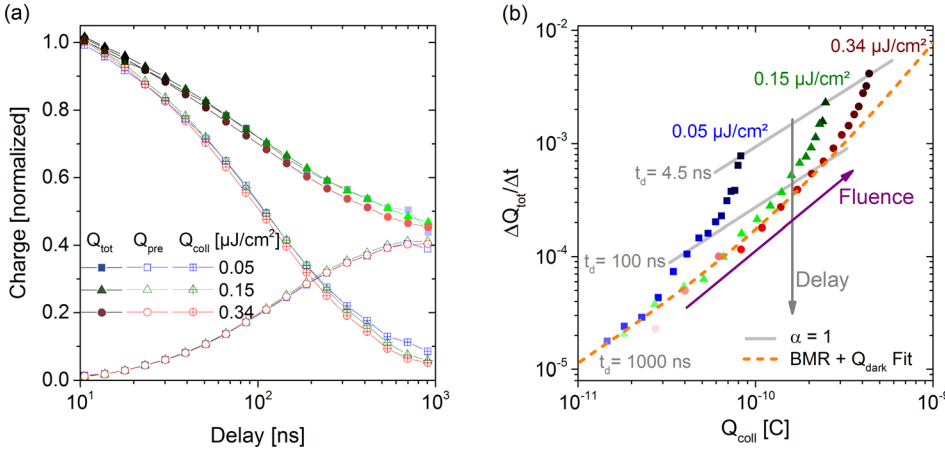


FIG. 2. (a) TDCF experiments with increasing delay measured for three different fluences; 0.05 (green), 0.15 (dark yellow), 0.34 (blue)  $\mu\text{J}/\text{cm}^2$  at  $V_{\text{pre}} = 0.3$  V and (b) a direct plot of the recombination order, which is two for longer delays and one (gray line) for all others. The fit (dashed orange line, Eq. (5), is done with  $k_2 = 1.5 \times 10^{-10} \text{ m}^3/\text{s}$  and  $Q_{\text{dark}} = 8 \times 10^{-11} \text{ C}$ .

increasing delay denotes only the loss of charges to the non-geminate recombination.

We performed the TDCF measurements at a forward bias where the injection is reasonably small compared to the photo-generated charge (e.g.,  $V_{\text{pre}} = 0.3$  V). The delay was continuously increased up to 2  $\mu\text{s}$ , at which  $Q_{\text{coll}}$  is no longer detectable and  $Q_{\text{tot}}$  reaches a constant value. Interestingly, the course of the normalized  $Q_{\text{tot}}$  as function of  $t_{\text{del}}$  is nearly independent of fluence, meaning that the non-geminate recombination after pulsed illumination initially does not depend on the charge density. This implies that the recombination follows a recombination order of one ( $\alpha = 1$ , see Figure 2(b), gray lines), which denotes a monomolecular decay process. This is in clear contrast to the results from the steady state BACE outlined above, which showed the non-geminate recombination (NGR) to be bimolecular under the steady state illumination conditions above 1 sun illumination intensity. It suggests that the initial decay of the photogenerated free carriers in TDCF may involve (initially empty) traps. The reported trap densities,  $N_{\text{T}}$ , in PbS QCD, are typically quite high, ranging between 10<sup>16</sup> and 10<sup>17</sup> depending on the type and density of the ligand.<sup>33</sup> A closer look at the TDCF data, however, reveals that the decay characteristics are inconsistent with a simple first order decay process. If trapping and recombination are strictly monomolecular processes with a time-independent first order recombination coefficient  $k_1$ , then the incremental reduction in  $Q_{\text{tot}}$  with the delay time would follow

$$\frac{[Q_{\text{coll}}(t + \Delta t) - Q_{\text{coll}}(t)] + [Q_{\text{pre}}(t + \Delta t) - Q_{\text{pre}}(t)]}{\Delta t} = \frac{\Delta Q_{\text{tot}}}{\Delta t} = -k_1 Q_{\text{coll}}. \quad (4)$$

This is, however, not consistent with the measured decay dynamics (Figure S5, red dashed line). To further understand the recombination order and dynamics in this system, an approach was adopted to analyze the non-geminate recombination loss, following the procedure outlined in Ref. 34. In short, the incremental change in  $Q_{\text{tot}}$  over a time increment,  $\Delta Q_{\text{tot}}(t)/\Delta t$ , is plotted versus  $Q_{\text{coll}}(t)$  in a log-log fashion. Note that  $\Delta Q_{\text{tot}}(t)$  is the loss due to NGR and  $Q_{\text{coll}}(t)$  is the charge remaining in the device at time  $t$ . Therefore, for any given delay time, the slope of  $\log(\Delta Q_{\text{tot}}(t)/\Delta t)$  versus  $\log(Q_{\text{coll}}(t))$  yields the order of NGR. In fact, at short delays, when  $Q_{\text{coll}}$  is

high, the slope of the line of best fit for the three fluences indeed gives  $\alpha = 1$  (Figure 2(b), gray line), which indicates a first order recombination process.

However, if the system followed a simple recombination process with a time-independent recombination coefficient, meaning that the recombination rate is an exclusive function of the charge density in the device, all points in the differential decay plot would fall on a single curve, which is clearly not the case (Figure 2(b)). At short delays, all three fluences deviate significantly from the single common curve traced out at longer delays.

A consistent explanation of both facts is that the NGR dynamics up to 100 ns is governed by a first-order process, which gradually slows down with time. Although we cannot be sure about the physical processes governing this phenomenon, we propose that the initial rapid decay is due to trapping of the faster carrier while trapping and recombination of the slower carrier dominates the dynamics at longer times. For even longer delays, all  $\Delta Q_{\text{tot}}(t)/\Delta t$  versus  $Q_{\text{coll}}(t)$  data finally approach one common curve, which now follows strict bimolecular recombination in presence of dark charge, according to Eq. (4)

$$\frac{\Delta Q_{\text{tot}}}{\Delta t} = \frac{1}{eAd} k_2 (Q_{\text{coll}}^2 + 2Q_{\text{coll}}Q_{\text{dark}}). \quad (5)$$

Here,  $8 \times 10^{-11} \text{ C}$  is the dark charge as determined by the BACE experiments on the same device in the dark. The value for  $k_2 = 1.5 \times 10^{-10} \text{ cm}^3/\text{s}$  from TDCF is slightly smaller than when determined by BACE. However, this is quite understandable considering the much lower carrier densities present in the device at long delays.

The data presented above reveal rather complex charge carrier dynamics in our PbS QD dot solar cells. While the properties under the steady state illumination at and above 1 sun (carrier density as a function of the illumination intensity,  $JV$ -characteristics) are consistent with the strict bimolecular recombination, the transient photocurrent data indicate the involvement of traps. A simple way to independently determine the order of recombination is to measure  $V_{\text{oc}}$  at different steady state intensities  $I$ , with the result shown in Figure S6. In this semilogarithmic presentation, the slope of the  $V_{\text{oc}} - \ln(I/I_0)$  plot is  $2k_{\text{B}}T/\alpha q$ , where  $\alpha$  is the order of the recombination process and  $k_{\text{B}}T/q$  is the thermal voltage. In the case of pure bimolecular recombination,  $\alpha = 2$ , while

$\alpha = 1$  for trap-assisted recombination. Consistent with results on the other CQD solar cells,<sup>35,36</sup> our measurements reveal two distinct ranges. While at low intensities the role of the traps is dominant,  $\alpha$  approaches a value of 2 above 1 sun illumination intensity. This indicates direct free carrier recombination dominates at the solar illumination conditions, in full agreement with the BACE results outlined above.

Our studies reveal a conclusive picture of the processes determining the steady state *JV*-curves of the PbS based CQD solar cells. Specifically, the time-delayed collection field experiments reveal a weak yet distinct field-dependence of generation, setting an upper limit to the achievable fill factor. The transient and low intensity steady state measurements highlight the role of traps, but the non-geminate recombination is found to be strictly bimolecular at application relevant steady state conditions, with a high bimolecular recombination coefficient of  $3 \times 10^{-10} \text{ cm}^3/\text{s}$ . As a consequence of the fast non-geminate recombination loss, the carrier densities at the open circuit condition are fairly low, going along with a large  $V_{\text{oc}}$  loss of ca. 0.75 V (with respect to a bandgap of 1.18 eV). On the other hand, the rather high effective mobility of ca.  $2 \times 10^{-3} \text{ cm}^2/\text{Vs}$ , measured under the steady state illumination, allows for fast carrier extraction and ensures a reasonably high fill factor, despite the fast non-geminate recombination.

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